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## Magnetocaloric effect in Dy based chromium oxides

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Rare-earth based chromium oxides have attracted substantial research attention over the years because their unusual properties, such as magnetocaloric effect (MCE) [1-5]. An interesting member of these materials is  $R\text{CrO}_4$ , belonging to a family of  $\text{ABO}_4$ -type oxides, where  $A$  is a rare-earth and  $B = \text{P, As, Cr, V}$ .  $R\text{CrO}_4$  compounds crystallize in zircon or monazite-type structure depending on the size of the trivalent rare-earth ion and the  $B$  element [1]. The  $R\text{CrO}_4$  compounds are of significance because of the outer shell configuration ( $3d^1 4s^0$ ) of the rare and unstable  $\text{Cr}^{5+}$  ion and the anomalous super-exchange formed by the zircon-type structure.  $R\text{CrO}_4$  with  $R = \text{La, Nd, Sm, Eu}$  and  $\text{Lu}$  behave antiferromagnetically (AFM), while the remaining oxides in the  $R\text{CrO}_4$  compound family are ferromagnetic (FM) [1-4]. The magnetic orders of  $R\text{CrO}_4$  are dependent on the structure, based on the variation in the  $\text{Cr-O-R}$  bond angles and the interatomic distances [1]. In these materials, the  $R$  and  $\text{Cr}$  moments order simultaneously at the same temperature and the  $\text{R}^{3+}$  and  $\text{Cr}^{5+}$  ions, both influence the MCE [4]. In the present work, the magnetic and MCE properties of sol-gel synthesized  $\text{DyCrO}_4$  rare-earth compound are studied. The as-synthesized sample was found to have an amorphous phase. Calcination of the amorphous powder at  $500^\circ\text{C}$  for 2 h, transformed the hydroxide into  $\text{DyCrO}_4$  oxide. Transmission electron microscopy (TEM) analysis of  $\text{DyCrO}_4$ , showed that the sample is agglomerated, and grain boundaries are indistinguishable. For both samples, the selected area electron diffraction (SAED) patterns confirmed the crystallinity, with the energy dispersive spectroscopy (EDS) verifying the pure elemental composition. The susceptibility as a function of temperature,  $\chi(T)$ , shows paramagnetic (PM) to FM transition for  $\text{DyCrO}_4$  at  $21.6 \pm 0.1$  K. The positive value of Curie-Weiss temperature,  $C_W$ , confirms the FM behaviour of  $\text{DyCrO}_4$  sample below the Curie temperature,  $T_C$ . The maximum entropy change,  $\Delta S_M(T, H)$ , is observed at about 21.96 K, close to the FM transition of the sample. This observation shows that  $\text{Dy}^{3+}$  and  $\text{Cr}^{5+}$  ions both influence the MCE, as  $\text{Dy}$  and  $\text{Cr}$  moments order simultaneously at the same temperature, with  $T_C = 21.6 \pm 0.1$  K [4]. The transition temperatures observed in the  $(T)$  curves are further confirmed with  $M(\mu_0, H)$  measurements. Further calcining the amorphous powder at  $900^\circ\text{C}$  resulted in the formation of  $\text{DyCrO}_3$  having orthorhombically distorted perovskite structures [5]. The MCE properties of the  $\text{DyCrO}_3$  [5] and  $\text{DyCrO}_4$  samples showed that both samples are good for MR application, with  $\text{DyCrO}_4$  showing more efficiency than  $\text{DyCrO}_3$  by have high maximum  $\Delta S_M(T, H)$  and rate of cooling power (RCP) values. The cause of the observed anomaly in magnetic transition and MCE will be discussed.

### References

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MSc

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