# Effect of concentration on the optical and solid state properties of ZnO thin films deposited by aqueous chemical growth (ACG)

S L Mammah<sup>1,2</sup>, F I Opara<sup>2</sup>, F B Sigalo<sup>2</sup>, S C Ezugwu<sup>3</sup>, F I Ezema<sup>3</sup>

<sup>1</sup>Department of Science Laboratory Technology, School of Applied Sciences, Rivers State Polytechnic Bori, Rivers State, Nigeria sylvestermammah@yahoo.com

2Department of Physics, Faculty of Science, Rivers State University of Science and Technology, Nkpolu, Port Harcourt, Rivers State, Nigeria

3Department of Physics and Astronomy, University of Nigeria, Nsukka, Enugu State, Nigeria fiezema@yahoo.com sabroze@yahoo.com

**Abstract:** Thin films of Zinc Oxide (ZnO) having different concentrations were deposited using the Aqueous Chemical Growth (ACG) method. The films were characterized using Rutherford Back Scattering (RBS) spectroscopy for chemical composition and thickness, X-Ray Diffraction (XRD) for crystallographic structure, a UV-VIS spectrophotometer for the analysis of the optical and solid state properties which include spectral absorbance, transmittance, reflectance, refractive index, direct band gap, real and imaginery dielectric constants, absorption and extinction coefficients and a photomicroscope for photomicrographs. The average deposited film thickness was 100nm. The results indicate that the values of all the optical and solid state properties investigated vary directly with concentration except transmittance which is the reverse. Thus, the optical and solid state properties of ZnO thin film deposited by the Acqueous Chemical Growth method can be tuned by deliberately controlling the concentration of the precursors for various optoelectronic applications including its application as absorber layer in solar cells.

#### 1. Introduction

Zinc Oxide is an amphoteric oxide which exists in the form of white powder (zinc white) at room temperature <sup>[1]</sup>. Though zinc oxide exists naturally in the earth crust (zincite), it is the synthetic ZnO produced in laboratories that are usually used in industries <sup>[2]</sup>.

Crystalline zinc oxide is thermochromic<sup>[3]</sup>. Apart from the ability of ZnO to react with acids, it also react with bases to produce soluble zincates <sup>[4-6]</sup>. Zinc oxide has high stability at room temperature and decomposes into zinc vapour and oxygen at about  $1975^{\circ}c^{[7]}$ . Though zinc oxide can exist in either of

<sup>&</sup>lt;sup>1</sup> To whom any correspondence should be addressed.

hexagonal wurtzite, cubic zinc blende and cubic (rock salt) structure, the hexagonal wurtzite structure is the most common at room temperature. The different polymorphs of ZnO do not possess the property of symmetry inversion. This and other properties such as ionic bonding account for the strong piezoelectricity of both hexagonal and zinc blende ZnO<sup>[8],[9]</sup>.

The hexagonal wurtzite structure has the point group 6mm (Hermann – Mauguin notation) or  $C_{6v}$ 

(schoenflies notation) and a space group of P6<sub>3</sub>me or  $C_{6v}^4$ . Its lattice constants are a = 3.25  $\stackrel{0}{A}$  and C =

5.2  $\stackrel{0}{A}$  [10].

The many applications to which ZnO can be put have made it to be a material for constant scientific study. The diverse applications of ZnO are derived from its unique properties which include non-toxicity, good electrical conductivity, high luminous transmittance, good substrate adherence, good optical behaviour and stability in plasma atmosphere <sup>[11], [12]</sup>

ZnO has been tested and found to behave as a semiconductor having a wide bulk direct band gap of about 3.37ev at room temperature<sup>[13]</sup> ZnO has been used for the fabrication of light emitting diodes, varistors, photo detectors, piezoelectric cantilever, gas sensors, buffer layer in solar cells and in photonic crystals<sup>[14-21]</sup>.

As a result of the strategic importance of ZnO to humanity, various deposition methods have been used for its production in the form of thin films. Among these methods are the chemical bath deposition (CBD) <sup>[22-26]</sup>, Successive Ionic Layer Adsorption and Reaction (SILAR) method <sup>[27, 28]</sup>, Spray pyrolysis method <sup>[29]</sup>, Electro deposition method <sup>[30, 31]</sup>. The ACG method is a novel method for the deposition of ZnO thin films; hence the literature on it is sparse.

Indeed Ezema<sup>[26]</sup> reported that "Thin films of ZnO have been used but its preparation techniques have been restricted to sputtering, vacuum evaporation, chemical vapour deposition, spray prolysis, molecular beam epitaxy, sol gel and pulse laser".

Among the advantages which the ACG method have over the other methods are simplicity, low cost, reproducibility, availability of materials, environmental friendliness (non toxicity of residue), non requirement of surfactant, templates and complexing agents, low temperature requirements, ability to produce nanostructures, high purity (absence of surfactants), suitability for large scale production etc.<sup>[32, 33]</sup>.

This paper reports the effect of concentration of precursors on the optical properties of ZnO thin films deposited using the ACG method with  $Zn(NO_3)_2$  6H<sub>2</sub>O as precursor material. The spectral analysis of the optical and solid state properties was carried out using a Unico UV-2102 PC spectrophotometer.

## **2** Experimental Details

ZnO thin films were deposited on clean glass slides using Aqueous Chemical Growth method. Equimolar concentrations of hydrated zinc nitrate  $(Zn(NO_3)_2, 6H_2O)$  and hexamine in 80ml of water were used as precursors

Hexamine was used to make the zinc nitrate alkaline. Three samples having different concentrations of the precursors were prepared as shown in table 1 below:

Table 1: Three samples of different concentrations of the precursors										
Sample of mass	Zn (NO <sub>3</sub> ) <sub>2</sub> .6H2O	Mass of	Conc. of Zn(NO <sub>3</sub> ) <sub>2</sub> .6H <sub>2</sub> O	Conc. Of Hexane	Volume of H <sub>2</sub> O					
muss		Hexane		IICAUIIC	1120					
C <sub>3</sub>	0.60g	0.030g	0.025M	0.025M	80.00ml					
C <sub>4</sub>	1.20g	0.060g	0.050M	0.050M	80.00ml					
C <sub>2</sub>	2.40g	0.112g	0.100M	0.100M	80.00ml					

Masses of samples were measured using an analytical microbalance. The masses of the components of each sample were put into a 100ml pyrex bottle together with 80ml of water. The content was properly mixed using a magnetic stirrer

The pyrex bottles were then tightly corked and carefully placed in an oven at a temperature of  $90.00^{\circ}$ C.

The average deposition time for the films was twelve hours

The chemical reactions which resulted in the crystallization of ZnO are

 $\begin{array}{l} C_{6}H_{12}N_{4}+6H_{2}O\rightarrow 6HCHO+4NH_{3}\\ NH_{3}+H2O\rightarrow NH_{4}^{+}+OH^{-}\\ 2OH_{-}+Zn^{2+}\rightarrow ZnO+H_{2}O^{[34]} \end{array}$ 

### **3** Results and discussion

The elemental composition of sample  $C_2$  was analyzed using Rutherford Backscattering (RBS) is as shown in figure 1. Judging from the film composition shown in the table 2 below, we conclude that the elements contained in sample  $C_2$  are Zn (0.050) and oxygen (0.950) while the glass substrate has the composition 0(0.500), Si(0.120), Ca (0.100), Al (0.100) and Na (0.180). This is summarised in table 2. The thickness of the film is given as 100nm.

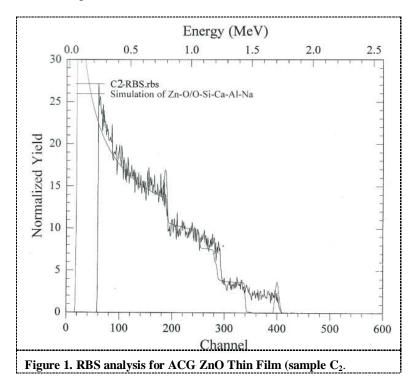
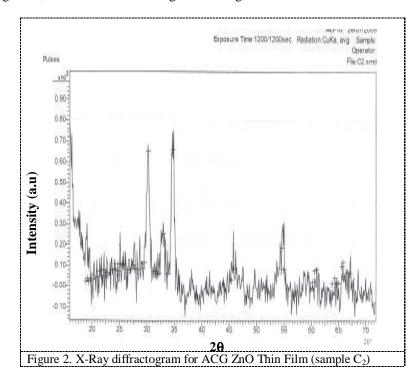


Table 1; Elemental composition of ZnO thin film and substrate from RBS analysis									
	Oxygen	Zinc	Silicon	Calcium	Aluminium	Sodium			
ZnO thin film	0.950	0.050	-	-	-	-			
Glass substrate	0.500	-	0.120	0.100	0.100	0.180			

The X-Ray diffractogram of the ACG ZnO thin films were studied to determine its crystalline nature.

The thin films were scanned continuously between O and 70 at step size of 0.03 and at time per step of 0.15s. The figure 2 above shows the intensity of peaks versus diffraction angle 2 $\Theta$  for sample C<sub>2</sub>Zn0 thin film using CuK<sub>a</sub> radiation source having a wavelength of 1.54056  $\stackrel{0}{A}$ .



The X-ray diffractogram reveal several peaks corresponding to directions of strong reflections. Some of the observed peaks are  $31.77^{\circ}$ ,  $34.84^{\circ}$ , and  $54.86^{\circ}$  having d values of 30.14, 34.42, and 54. 86 respectively.

The 34.84<sup>0</sup> agree with the preferred orientation along the (002) plane reported by Ajuba et al<sup>[35]</sup> and Shinde et al<sup>[36]</sup> who used the CBD and SILAR methods respectively.

The diffractogram reveal sharp diffraction peaks which is indicative of good crystallinity<sup>[37-41]</sup>.

The XRD pattern shows that film is crystallized in the wurtzite ZnO hexagonal structure (JCPDF CARD no. 36-1451)<sup>[42]</sup>.

The mean size of the crystallite was estimated to be 12nm using the Sherrer formula which is given as:-

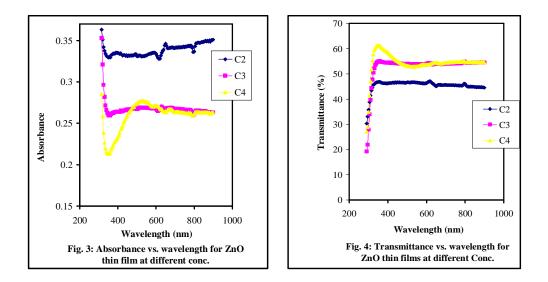
$$D = \frac{K\lambda}{\beta\cos\theta}$$

Where k = 0.9

 $\lambda = 1.541 \overset{0}{A}$ 

 $\theta$  is the diffraction peak angle (34.42<sup>0</sup>) and  $\beta$  is the Full Width at Half Maximum (FWHM) corresponding to the diffraction peak.

The spectral absorbance of the films is shown in fig. 3. Absorbance increases with concentration. The 0.1M concentration has a fairly constant absorbance value of about 0.34 and 0.35 at the visible and infrared regions respectively.



The 0.05M concentration has an average absorbance value of 0.28 in the visible region while the 0.025M concentration has an average absorbance value of 0.268 in the visible region.

However, the 0.05M concentration has the lowest absorbance value of 0.21 in the infrared region.

Absorption peaks clearly occurred in film  $C_2$  at 368nm, 496nm and 656nm but were not so clearly defined in films  $C_3$  and  $C_4$ . Ezema <sup>[26]</sup> reported absorption peaks at 3.68nm, 449nm and 566nm using Chemical Bath Method. The difference in the absorption peaks may be due to deposition method. The spectral transmittance of the films is as shown in fig.4. Transmittance decreases with concentration in the visible region. The 0.1M, 0.05M and 0.025M concentrations has average transmittance of 45.5%, 53.2% and 54.60% respectively in the visible region.

However, the 0.05M concentration has the highest transmittance of about 61.8% in the ultraviolet region. The transmittance of the films in the visible region is generally lower than the above 85% transmittance reported using Chemical Bath Deposition method<sup>[43]</sup>.

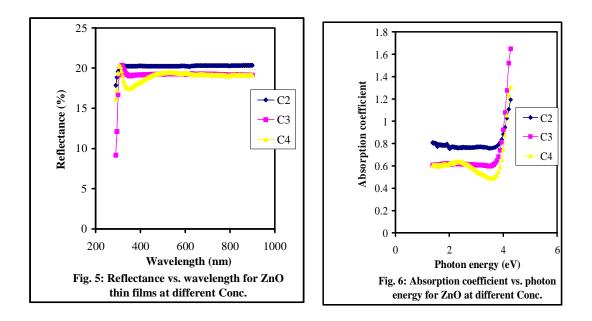
It is also much lower than the 90-95% transmittance in the UV-VIS-NIR regions for ZnO thin films deposited using spray pyrolysis method<sup>[44, 45]</sup> and Sol. Gel method<sup>[46]</sup>.

It has been reported that the transmittance of a film increases as its thickness decreases<sup>[26]</sup>. The average thickness of the films being reported on is 100nm which is far smaller than the 0.203µm and 0.069µm thicknesses which gave average transmittances of 62% and 75% respectively that has been reported. <sup>[26]</sup> Thus, the transmittance of films deposited by ACG method are relatively lower compared to the ones deposited by other methods even though the thicknesses are smaller.

The spectral reflectance of the films being reported on is as shown in fig. 5. Reflectance increases with concentration in the visible region.

The 0.1M, 0.050M and 0.025 concentrations reflect an average of 20.25%, 19.38% and 18.88% of visible electromagnetic waves. These values are higher than the average of 12 to 17% reflectance of films deposited using Chemical Bath Method as reported by Ezema<sup>[26]</sup>.

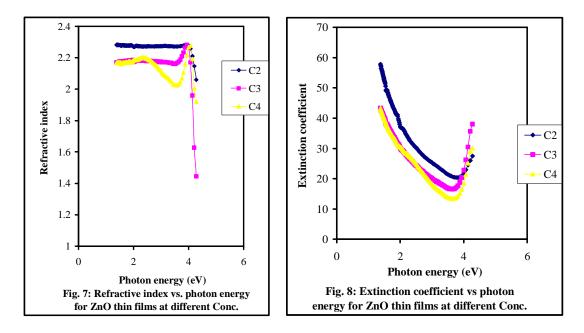
Thus, films deposited by ACG method are better reflectors than those deposited by Chemical Bath Method. Absorption coefficient increases with concentration in the visible region as shown in figure 6. The average absorption coefficients for the 0.025M, 0.05M and 0.1M concentrations are 0.62, 0.65 and 0.77 respectively. However, in the UV region, 0.05M has the lowest absorption coefficient of about 0.49.



Refractive index is fairly constant in the visible region as shown in figure 7. The 0.1M, 0.05M and 0.025M concentrations have average refractive indices of 2.3, 2.225 and 2.2 respectively. However, the 0.05M concentration has the lowest refractive index of about 2.05 in the UV region.

It has been reported that the refractive index of ZnO films doped with Li deposited by spray pyrolysis lie between 1.60 and 2.20 at a wavelength of 500nm<sup>[45]</sup>, while Ezema<sup>[26]</sup> gave the average refractive index of the ZnO films deposited by Chemical Bath method as being between 1.64 and 1.98 and also reported observed peak values of 2.28 at 368nm and 1.72 at 569nm.In our own case, the refractive indices for the different concentrations were generally uniform in the visible region.

Extinction coefficient decreases with concentration and also decreases with increasing photon energy (wavelength) in the visible region as shown in figure 8.



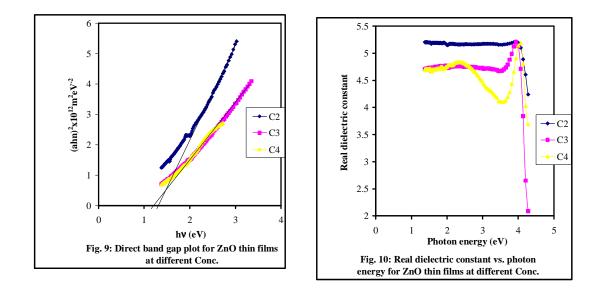
The value of extinction coefficient for 0.1M concentration ranges from  $37.85 \times 10^{-3}$  for 2eV to  $27.14 \times 10^{-3}$  for 3eV in the visible region. Similarly, the values of extinction coefficient for 0.05M concentration ranges from  $35 \times 10^{-3}$  for 2eV to  $19.64 \times 10^{-3}$  for 3eV and that of the 0.025M concentration ranges from  $30 \times 10^{-3}$  to about  $21.43 \times 10^{-3}$  at 3eV in the visible region

However, the 0.05M concentration has the lowest extinction coefficient of about  $13.93 \times 10^{-3}$  in the ultraviolet region. In general all the concentrations have the lowest value of extinction coefficient in the higher energy region (between 3eV and 4eV) and the maximum value in the low energy region (between 1eV and 2eV). Low and high values of extinction coefficient at high and low energy regions respectively have been reported<sup>[26]</sup>.

The band gap increases with concentration as shown in figure 9. The average band gap for the 0.025M, 0.05M and 0.1M concentrations are 1.16eV, 1.22eV and 1.4eV respectively.

These values are lower than the direct band gaps of ZnO films deposited by Chemical Bath method<sup>[26]</sup> which lie between 1.60eV and 1.80eV.

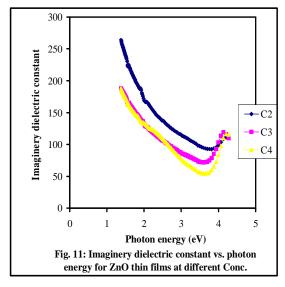
Our values are also far lower than 3.29eV band gap obtained for aluminium doped ZnO deposited using radio frequency (r.f) megnetron sputtering <sup>[18]</sup>, and intrinsic band gap of 3.20eV for ZnO films deposited by spray pyrolysis<sup>[29]</sup>. The low values of the direct band gap obtained indicate that the ZnO thin film materials prepared by ACG method are suitable for use as absorber layers in solar cells. The low values of the band gaps obtained in this work may be due to preparation conditions Ezema et al<sup>[26]</sup>. Real dielectric constant increases with concentration in the visible region as shown in figure 10. The 0.025M, 0.05M and 0.1M concentrations have average real dielectric constants of 4.76, 4.85 and 5.15 respectively in the visible region of the electromagnetic spectrum.



However, the 0.05M concentration has the lowest real dielectric constant of about 4.09 in the ultra violet region. Figure 11 is the Imaginary dielectric constant that increases with concentration in the visible range.

The 0.1M, 0.05M and 0.025M concentrations have average values of imaginary dielectric constant which ranges from  $180.3 \times 10^{-3}$  for 2eV to  $118.18 \times 10^{-3}$  for 3eV,  $143.93 \times 10^{-3}$  for 2eV to  $78.785 \times 10^{-3}$  for 3eV and  $143.93 \times 10^{-3}$  for 2eV to  $89.39 \times 10^{-3}$  for 3eV respectively. This shows that the imaginary dielectric constant decreases with increasing photon energy (decreasing wavelength) for all concentrations in the visible region.

However, the 0.05M concentration has the lowest imaginary dielectric constant of about 56.06 in the ultra violet region. All concentrations have their lowest imaginary dielectric constant values in the ultra-violet region.



The photomicrographs as shown in figure 12 indicate uniform deposition of the ZnO thin films on the glass substrate. While C2 and C3 show stretches of rod C4 shows clusters of thin films on the substrates .



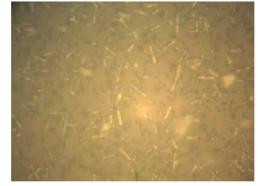






Fig. 12: Photomicrographs of ACG ZnO thin films

#### 4. Conclusion

Crystals of ZnO have been successfully grown on glass slides in the form of thin films from the aqueous solution of hexahydrated zinc nitrate and hexamine after the fashion of Lionel Vassieres<sup>[47]</sup>. The concentration of the precursors was found to vary directly with absorbance, reflectance, absorption coefficient, extinction coefficient, refractive index, direct band gap, real dielectric constant and imaginery dielectric constant. The transmittance of the thin films was found to vary inversely with the concentration of the precursors. The results indicate the suitability of ZnO thin films prepared by the Aqueous Chemical Growth method for various optoelectronic applications such as absorber layer in solar cells.

#### **REFERENCES:**

- 1. Takahashi K, Yoshikawa A, Sandhu A 2007. Wide bandgap semiconductors: fundamental properties and modern photonic and electronic devices. (Springer.) p. 257.
- 2. Klingshirn C, 2007. Chem. PhysChem 8 (6): 782.
- 3. Wiberg, E and Holleman, A F 2001. Inorganic Chemistry. Elsevier.
- 4. Nicholson J W, 1998. Journal of materials Science 33: 2251
- 5. Ferracane, Jack L. 2001. Material in Dentistry: Principles and Application. (Lippincott :Williams & Wilkins.) Pp. 70, 143.
- 6. Park C K, Silsbee M R., Roy D M, 1998. Cement and concrete research 28 (1): 141-150.
- 7. Greenwood, N N and Earnshaw, A 1997. Chemistry of the Elements. (Oxford: Butterworth-Heinemann.)
- 8. Özgür U, Alivov Ya. I., Liu C, Teke A, Reshchikov, M. A.; Doğan, S.; Avrutin, V.; Cho S.J et al. 2005. Journal of Applied Physics 98: 041301.
- 9. Baruah S and Dutta J 2009, Sci. Technol. Adv. Mater 10.013001.
- 10. Rossler, U, 1999. Landolt-Bornstein, new Series, Group III vol. 17B ed., 22, 41B. (Springer, Heidelberg.)
- Mahalingam T, John V S, Raja M, So Y K, Sabastian P J, 2005 (Sol. Energy Mater. Sol. Cells.) pp88-227
- 12. Shinde Y R, Gujar T P, Lokhande C D, Nane R S, Han S H 2006 Mater. Chem. Phys. p326
- 13. Ajuba A E, Ezugwu S C, Asogwu P U, Ezema F I 2010 (Chalcogenide letter vol 7) p573-579
- 14. Soki T, Hatanaka Y, Look D C 2000 (Appl. Phys. Lett. vol. 76) p3257
- Lin Y, Gorla C R, Linng S, Emanetoglu N, Tor Y, Shen H, Wraback M, 2000 (J. Electron Mater. Vol. 29) p60
- 16. Shinde V R, Gujar T P, Lokhande C D 2007 (Sens. Actuators H. vol. 20) p551
- Ennaoui A, Siebentrith S, Lux-Steiner M C, Riedl W, Karg F 2002 (Sol. Energy Mater. Sol. Cells vol. 73) p51

- Chen Y, Bagnall D, Yao T, Mater 2000 (Scien Eng., B, solid-state mater, Adv. Technol, vol. 75) p190.
- 19. Liang S, Sheng H, Liu Y, Hio Z, Lu Y, Shen H 2001 (J. Cryst. Growth vol.225) p110
- 20. Koch M H, Timbrell P Y, Lamb R N 1995 (Semicond. Sci. technol. Vol.10) p7523
- Gorla C R Emanetoglu N W, Liang S, Mayo W E, Lu K, Wraback M, Shen H, 1999 (J. Appl. Phys. Vol. 85) p2595
- 22. Shinde V R, Lokhande C D, Mane R S, Han S H 2005 (Appl. Surf. Sci. vol 245) p407
- Ennaoui A, Weber M, Scheer R, Lewerenz H 1998 (J Sol. Energy Mater. S. Cells vol 54) p277
- 24. Boyle D S, Governder K, O'Brien P 2002 (Chem. Commun.) p80
- Ortega-Lope M, Avila-Gaecia A, Albor-Aguitera M L, Sankez Resendiz V M 2003 (Mater. Res. Ball. Vol 38) p1241
- 26. Ezema F I 2004 Journal of Research Science Vol. 15 pp 343-350
- 27. Shinde V R, Gujar T P, Lokhande C D 2007 Solar Energy Materials and Solar Cells vol 91 p1961
- 28. Jimenez-Gonzalez A, Suarez- Para R. 1996 Cryst. Growth vol 167 p649
- 29. De la Olvera M L, Maldonado Asomoza A R, Melendez Lira M, 2002 Sol. Energy Mater. Sol. Cells vol 71
- 30. Izaki P M, Omi T, 1997 J. Electrochem Soc. Vol 144 p1949
- 31. Gal D, Hodes G, Lincot D, Sechock H W 2000 Thin Solid films vol. 362 p79
- 32. Xiulan H, Yoshitake M, Tatsuki O, Kazimi K 2009 Thin solid films: Elsevier vol. 518 pp621-624
- 33. Zhang X, Wang L and Zhow G 2005 Rev. Adv. Mater Sci vol. 10 pp60-72
- Shih-Fong L, Li-Ying L, Yung-Ping C Journal of Science and Engineering Technology, vol. 5 No. 3 pp 13 - 20
- Ajuba A E, Ezugwu S C, Ezekoye B A, Ezema F I , Asogwa P U 2010 Journal of Optoelectronics and Biomedical Materials, Vol. 2 issue 2. p73-78
- 36. Shinde V R, Lokhande C D, Mane R S, Han S H 2005 Appl. Surf. Sci. vol. 245 p407
- 37. Ping L, Hui L, Fan-Xiang X, Yu W 2008 Materials Chemistry and Physics1vol.12 p393
- Hong-Ju Z, Wei-hong W, Fei L, Hai-Shui W 2008 Material Chemistry and Physics pp112 1024

- Doungporn Y, Kanittha B, Wiyong K, Wan S 2009 Journal of microscopy society of Thailand, vol.13 p75
- 40. Sheng-Yuan C, Tser-Min Y 2000 Journal of Materials Science Letters vol.19 p349
- Geeth D, Tilagarathi T 2010 Digest Journal of nanomaterials and Biostructures, Vol. 5, No.1 p.297-301
- 42. Gumus C, Ozkendir O M, Kavak H, Ufuktepe Y 2006 Journal of optoelectronics and Advanced materials Vol. 8. No.1 pp299-303
- 43. Cruz-Vazquez A, Rocha-Alonzo F, Burruel-Ibarra S E, Inoue M and Bernal R 2001 Superficies y Vacio vol. 13, 89-91
- 44. Sachez-Juarez A, Tiburcio-Silver A and Ortiz A 1998 Solar Energy Materials and Solar Cells, vol. 52 pp301-311
- 45. Pushparajah, P, Arof A K and Radhakrishna S 1994 J Phys. D vol. 27 pp1518-1521
- 46. Jimenez-Gonzalex, A E and Soto-Urueta, J A 1998 Solar energy Materials and Solar Cells, vol. 52 pp345-353
- 47. Vayssieres L 2004 Int. J. of Nanotechnology Vol. 1